



# Eco-friendly controllable synthesis of highly dispersed ZIF-8 embedded in porous Al<sub>2</sub>O<sub>3</sub> and its hydrogenation properties after encapsulating Pt nanoparticles

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## ABSTRACT

In this paper, a new strategy was proposed to synthesize a novel composite of highly dispersed ZIF-8 embedded in porous Al<sub>2</sub>O<sub>3</sub> by a solvent-free method. This is a very simple, efficient, and eco-friendly method, in which ZnO/Al<sub>2</sub>O<sub>3</sub> was used as a precursor, and dimethyl imidazole was used as an organic ligand. Meso- and microporous composite of ZIF-8@Al<sub>2</sub>O<sub>3</sub> was synthesized arising from the crystal growth of ZIF-8 derived from ZnO nanoparticles. Characterization results reveal that the distribution of micropores and mesopores can be controlled by simply adjusting the amount of zinc salt supported on Al<sub>2</sub>O<sub>3</sub>. The value of this composite is that it can simultaneously exert the advantages of ZIF-8 and Al<sub>2</sub>O<sub>3</sub> support, which can significantly decrease the particle size of ZIF-8 and improve its utilization efficiency. The micropores of ZIF-8 can increase the dispersion of Pt nanoparticles, at the same time, the functional groups on the surface of Al<sub>2</sub>O<sub>3</sub> can improve the selectivity of the product. Furthermore, this kind of composite material can be extended by replacing the support. Finally, the results of the activity tests showed that the obtained Pt/ZIF-8@Al<sub>2</sub>O<sub>3</sub> catalyst exhibits high activity and selectivity in hydrodearomatization (HDA) and hydrodeoxygenation (HDO) reactions.

## 1. Introduction

Metal organic frameworks (MOFs) as a novel class of microporous materials, have been extensively studied because of their crystalline ordered networks and high surface area. MOFs have great potential applications in gas storage and separation, drug delivery, and especially in catalysis [1–5]. For more applications, lots of efforts have been focused on synthesizing MOF-based composites to enhance electrical [6–8], adsorption [9] and thermal properties [10–13]. For examples, Zhang et al. reported a hybrid material with enhanced electrocatalytic ability through confining a Cu-based MOF in macroporous carbon [14]. Recently, Pourebrahimi et al. also have synthesized MIL-101(Cr)/GNP composite by embedding graphene nanoplates into MIL-101 pores, and the composite showed an enhanced CO<sub>2</sub> adsorption capacity. [15] These works all have expanded ways for application or enhanced properties of MOFs.

In recent years, applying MOFs on heterogeneous catalysis has received particular interests not only because it possesses a high area surface, but it also opens up a new opportunity in green catalysis

[16–22]. However, narrow pores of MOFs limited the diffusion and passing through of large molecules, which makes it cannot compete with mesoporous materials for many reactions. Hence, meso-microporous composite materials based on composting MOFs with inorganic mesoporous materials have been reported to circumvent this limitation [23,24]. Gorka et al. and coworkers reported a one-pot microwave hydrothermal method for synthesis of MOF-boehmite and silica composites using triblock-copolymer [25]. Furtado et al. reported a composite material composed of an inorganic silica modified with CuBTC [26]. Recently, a modulated formation of MOF-5 by oriented growth over SBA-15 was reported by Karimi et al., and this research gives a new strategy for designing heterogeneous catalyst based on MOFs [27]. Although above composite materials indeed offered better properties compared to the pristine MOFs, but most approaches required pre-functionalization of the inorganic matrix to attract a growth of MOFs crystals, which makes the preparation more complicated. Besides, there are some impurities may exist in the hybrid composite [15], and it is hard to control the ratio of two components [25]. Therefore, to develop a simple, efficient, and eco-friendly strategy for synthesizing MOFs

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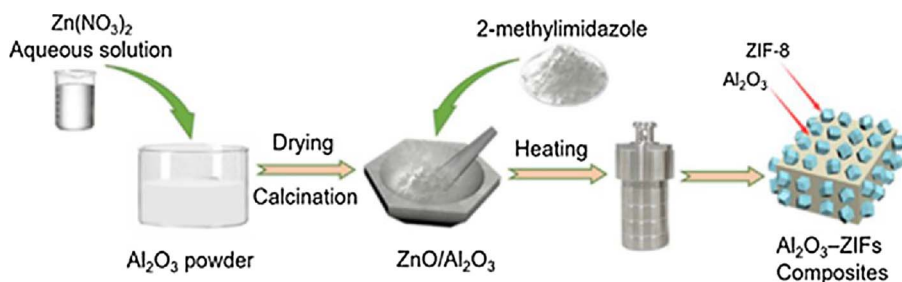
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Fig. 1. Synthetic scheme of ZIF-8@Al<sub>2</sub>O<sub>3</sub> composites.

composites with controllable meso- and microporosity, proper acidity, and surface area is still a great challenge.

As is well known, ZIF-8 is a famous ZIF compounds because of its relatively high thermal stability, high porosity, and large surface area, which has shown great potential in various research and industrial areas, especially in hydrogenation catalysis [28–31]. Benefit from stability and inherent acidity [32–35], many inorganic materials have been widely used as catalyst supports. Aiming at combining high surface area of ZIF-8 with proper acidity of inorganic support materials, we describe herein a controllable synthetic process of highly dispersed ZIF-8 embedded in porous Al<sub>2</sub>O<sub>3</sub> by a solvent-free method. Comparing with an extensive fabrication process involving wet-chemistry-based crystallization and separation, this method is highly efficient, controllable, and environment friendly as well as applicable for scale-up production and device fabrication. To our knowledge, this is the first report of introducing solid state method [36] to fabricate ZIF-8 based composite.

In this paper, we proposed a new strategy to synthesize the composite of highly dispersed ZIF-8 embedded in porous Al<sub>2</sub>O<sub>3</sub> by a solvent-free method. As illustrated in Fig. 1, this procedure is a universal and high-yield strategy for porous materials compositing with ZIF-8, and the component can be controlled by simply adjusting the impregnation amount of zinc salt. A series of ZIF-8@Al<sub>2</sub>O<sub>3</sub> composites with different amounts of ZIF-8 were synthesized. The morphology, component, texture, and acid property of composites were all carefully characterized and analysed. Additionally, Pt/ZIF-8@Al<sub>2</sub>O<sub>3</sub> catalyst was synthesized and tested using hydrogenation reaction, which exhibits high activity and selectivity in HDA and HDO reactions.

## 2. Experimental

Al<sub>2</sub>O<sub>3</sub> support was purchased from Tianjin Chemist Scientific Ltd. China and calcined at 600 °C for 2 h to transform into  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> phase before use. All other materials and reagents used in this work are also commercially available and were used as received without further purification.

### 2.1. Synthesis of composites and catalysts

Firstly, Bulk ZIF-8 was synthesized by a solid state method. ZnO powder with a diversity of morphologies were successfully prepared with the addition of different surfactants by a simple low-temperature liquid-phase method. The synthesized process of bulk ZIF-8 was according to previously reported studies [36,37], 2-methylimidazole was mixed with the synthesized ZnO powder at a molar ratio of 2.2: 1 (2-methylimidazole: ZnO), then the mixture was grinded in a mortar and sealed in an autoclave, followed by heating at 180 °C for 18 h. Finally, the product was obtained as a white powder and washed three times with deionized water, and dried in a vacuum oven at 120 °C for 4 h.

ZIF-8@Al<sub>2</sub>O<sub>3</sub> was also prepared using a solvent-free solid state method. The procedure was illustrated in Fig. 1. Firstly, Zn(NO<sub>3</sub>)<sub>2</sub>/Al<sub>2</sub>O<sub>3</sub> was prepared from Al<sub>2</sub>O<sub>3</sub> by incipient wetness impregnation of zinc nitrate hexahydrate with different loading. Then ZnO/Al<sub>2</sub>O<sub>3</sub> was obtained after calcined at 350 °C for 2 h. Subsequently, the supported oxide precursor was mixed with 2-methylimidazole at a molar ratio of

5: 1 (2-methylimidazole: ZnO). Then the mixture was grinded in a mortar and sealed in an autoclave, followed by heating at 180 °C for 18 h. Finally, the product was obtained as a white powder and was washed three times with deionized water, and dry in a vacuum oven at 120 °C for 4 h.

0.7 wt% Pt/ZIF-8@Al<sub>2</sub>O<sub>3</sub> catalyst was synthesized by incipient wetness impregnation method. Because ZIF-8 is hydrophobic (test result is shown in Fig. S1) and low acid-resistant, acetone was chosen as solvent in this process. In a typical process, acetylacetonate platinum was dissolved in acetone and then the solution was impregnated into ZIF-8@Al<sub>2</sub>O<sub>3</sub> support. The wet sample was dried at 80 °C for 6 h, and then reduced at 350 °C for 2 h in a flow H<sub>2</sub> atmosphere (50 mL/min) to obtain supported Pt nanoparticles. The reduced Pt/ZIF-8@Al<sub>2</sub>O<sub>3</sub> catalysts were all passivated in a N<sub>2</sub> atmosphere at 40 °C for 3 h, before being exposed to air. For comparison, Pt/Al<sub>2</sub>O<sub>3</sub> and Pt/ZIF-8 catalysts were also synthesized using the same impregnation method as above. In this paper, the Pt loading of all catalysts was 0.7 wt%.

### 2.2. Characterizations

Powder X-ray diffraction (XRD) patterns of samples were recorded on Bruker D8 focus diffractometer, with Cu K $\alpha$  radiation at 40 kV and 40 mA. Transmission electron microscopy (TEM) images were obtained using FEI Tecnai G2 F20 field emission microscopy at an accelerating voltage of 200 kV. X-ray photoelectron spectroscopy (XPS) was tested on Kratos Analytical Ltd—Axis Ultra DLD spectrometer equipped with an Al K $\alpha$  X-ray source (250 W). And the test was carried out with an analyzer pass energy of 188 eV for survey scans and 30 eV for detailed elemental scans. The Brunauer-Emmett-Teller (BET) surface area and pore volume were obtained by using nitrogen adsorption-desorption isotherms at 77 K with DKSH Belsorp-Max equipment. Prior to nitrogen sorption measurement, all samples were pretreated at 150 °C for 10 h. Acid feature of samples was analyzed by temperature programmed desorption of ammonia (NH<sub>3</sub>-TPD) and the reduction capacity of samples were analyzed by temperature programmed reduction of H<sub>2</sub> (H<sub>2</sub>-TPR) experiment, which were both performed on Micromeritics Chemisorb 2750 gas-adsorption instrument. The sample (about 200 mg) was placed in quartz reactor and pretreated in N<sub>2</sub> at 200 °C for 2 h, then calmed to room temperature at N<sub>2</sub> atmosphere. Subsequently, for NH<sub>3</sub>-TPD test, dose 10% NH<sub>3</sub>/He onto sample for 30 min then the sample was purged with pure He at 60 °C for 30 min to remove physical adsorbed NH<sub>3</sub>, and then heated to 800 °C with a rate of 3 °C/min. For H<sub>2</sub>-TPR test, when the pretreated sample was calmed at room temperature, the N<sub>2</sub> was switched to a 10% H<sub>2</sub>/Ar. Then the sample was heated to 800 °C with a rate of 10 °C/min. Thermogravimetric (TG) analysis was carried out with materials heated in nitrogen atmosphere at a ramp rate of 10 °C/min to 900 °C.

### 2.3. Catalytic performance test

Catalytic reaction was performed on a continuous-flow fixed-bed reactor. The catalysts were all pelleted, crushed, and sieved with 20–40 mesh before using for hydrogenation reaction. The react material for HDA was 5 wt% naphthalene dissolved in cyclohexane, and for HDO

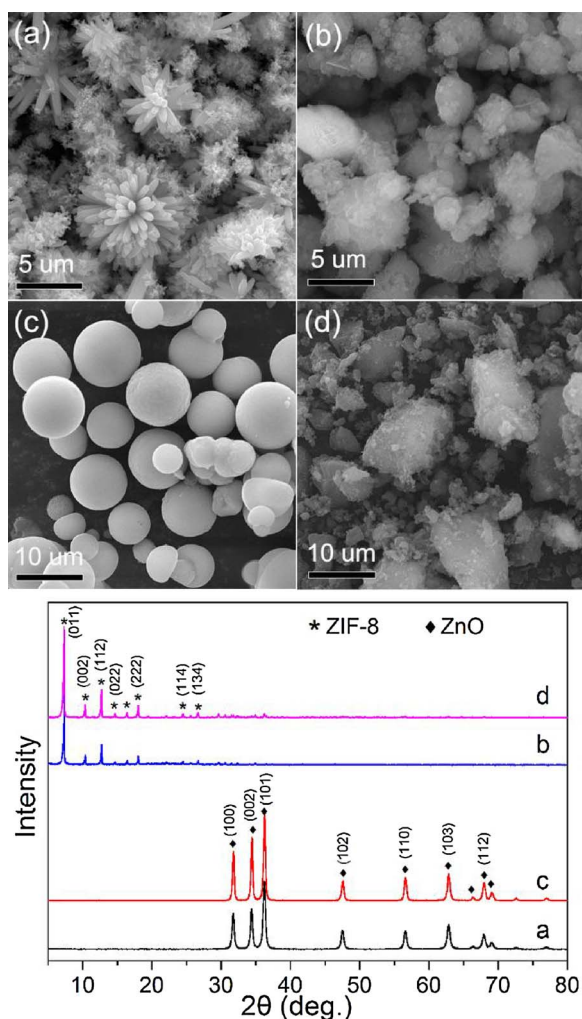


Fig. 2. SEM images and corresponding XRD patterns of (a) flower-shaped ZnO, (b) ZIF-8 synthesized from (a), (c) spherical ZnO, (d) ZIF-8 synthesized from (c).

was 50 wt% methyl palmitate dissolved in decalin. The test conditions were set at 3 MPa, WHSV = 1 h<sup>-1</sup>, and H<sub>2</sub>/oil = 600. Liquid products were collected and analysed with an Agilent 7890A/5975C GC-MS equipped with a HP-5 column.

### 3. Results and discussion

#### 3.1. Synthesis and structural characterization

In order to express the composition and content of the composites more clearly, the composite was denoted as ZA-n (ZIF-8@Al<sub>2</sub>O<sub>3</sub>) with n represent different mass percent of Al<sub>2</sub>O<sub>3</sub>.

In order to observe the changes in the morphology of the reactant and product after the solid state reaction, bulk ZIF-8 samples (Fig. 2b, d) were synthesized from flower-shaped (Fig. 2a), and spherical (Fig. 2c) ZnO, respectively. The ZIF-8 crystals with irregular shape were successfully synthesized, which morphologies are entirely different from those of ZnO precursors. The morphology of ZnO was destroyed totally due to a corrosion of 2-methylimidazole at 180 °C and crystal was reconstructed during reaction. This phenomenon also appeared on ZIF-8@Al<sub>2</sub>O<sub>3</sub> composites which will be shown in the later of this paper.

Fig. 3a shows the XRD patterns of ZIF-8, Al<sub>2</sub>O<sub>3</sub>, and ZIF-8@Al<sub>2</sub>O<sub>3</sub> with different percentage of Al<sub>2</sub>O<sub>3</sub>. The diffraction peaks of ZIF-8 all match well with literature [38]. The two diffraction peaks at 46° and 67° are attributed to γ-Al<sub>2</sub>O<sub>3</sub> phase (JCPDF: 1-1308), which were detected on Al<sub>2</sub>O<sub>3</sub> and ZA-n (n = 90, 80, 70, 60). Major diffraction peaks

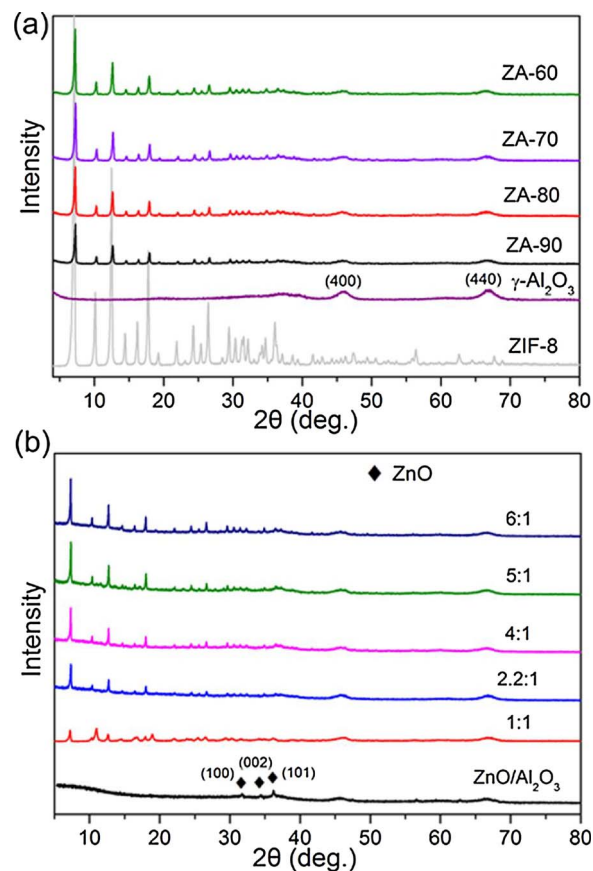


Fig. 3. XRD patterns of (a) ZIF-8@Al<sub>2</sub>O<sub>3</sub> with different proportion of Al<sub>2</sub>O<sub>3</sub>, (b) the synthesized ZA-70 with different molar ratios (2-methylimidazole: ZnO).

of ZA-n are identical with that of ZIF-8 phase, emphasizing successful preparation of ZIF-8 incorporation into matrix Al<sub>2</sub>O<sub>3</sub>. As we can see, compared with ZIF-8, the intensity of the main diffraction peak in ZA-n is obviously reduced, which is more obvious with the increase of Al<sub>2</sub>O<sub>3</sub> content. The intensity decline and shape broadening of diffraction peaks may be due to the reduced size of the ZIF-8 crystal particles in ZA-n composites.

It is well known that the molar ratio of organic ligand and metal salt has a remarkable effect on the synthesis of MOFs. To study the effect of the molar ratio of organic ligand and metal salt on the product, various samples were synthesized at different molar ratios using Al<sub>2</sub>O<sub>3</sub> as support (Fig. 3b). The precursor of ZnO/Al<sub>2</sub>O<sub>3</sub> (70 wt% Al<sub>2</sub>O<sub>3</sub>) has only three weak diffraction peaks (at about 2θ = 32°, 34°, and 36°) attributed to hexagonal ZnO structure (JCPDF: 1-1136). The XRD results of composites show that diffraction peaks of ZIF-8 were detected when the molar ratio was 2.2: 1, but weak diffraction peaks of ZnO were also detected. The intensity of diffraction peaks of ZIF-8 reaches its maximum value when the molar ratio was 5: 1, and no obvious peaks of ZnO was detected. The above results indicate that the synthesis of ZA-n composite requires a higher molar ratio than that of pure ZIF-8, which might be explained by that ZnO was in pores of Al<sub>2</sub>O<sub>3</sub> support and cannot fully contact and react with 2-methylimidazole when the molar ratio was small.

Fig. 4 shows the SEM images of ZIF-8, Al<sub>2</sub>O<sub>3</sub>, and ZA-n. Similar to the previously published literature [37], the ZIF-8 synthesized using solid phase method does not have a specific morphology and particle size (Fig. 4a). Only a few particles can be seen as a rhombic dodecahedron structure. For alumina, it has an amorphous structure aggregated with small particles (Fig. 4b). The crystal structure of ZIF-8 could not be observed from SEM images in the ZA-n (Fig. 4c–e), which might be caused by ZIF-8 component was embedded in Al<sub>2</sub>O<sub>3</sub> support.



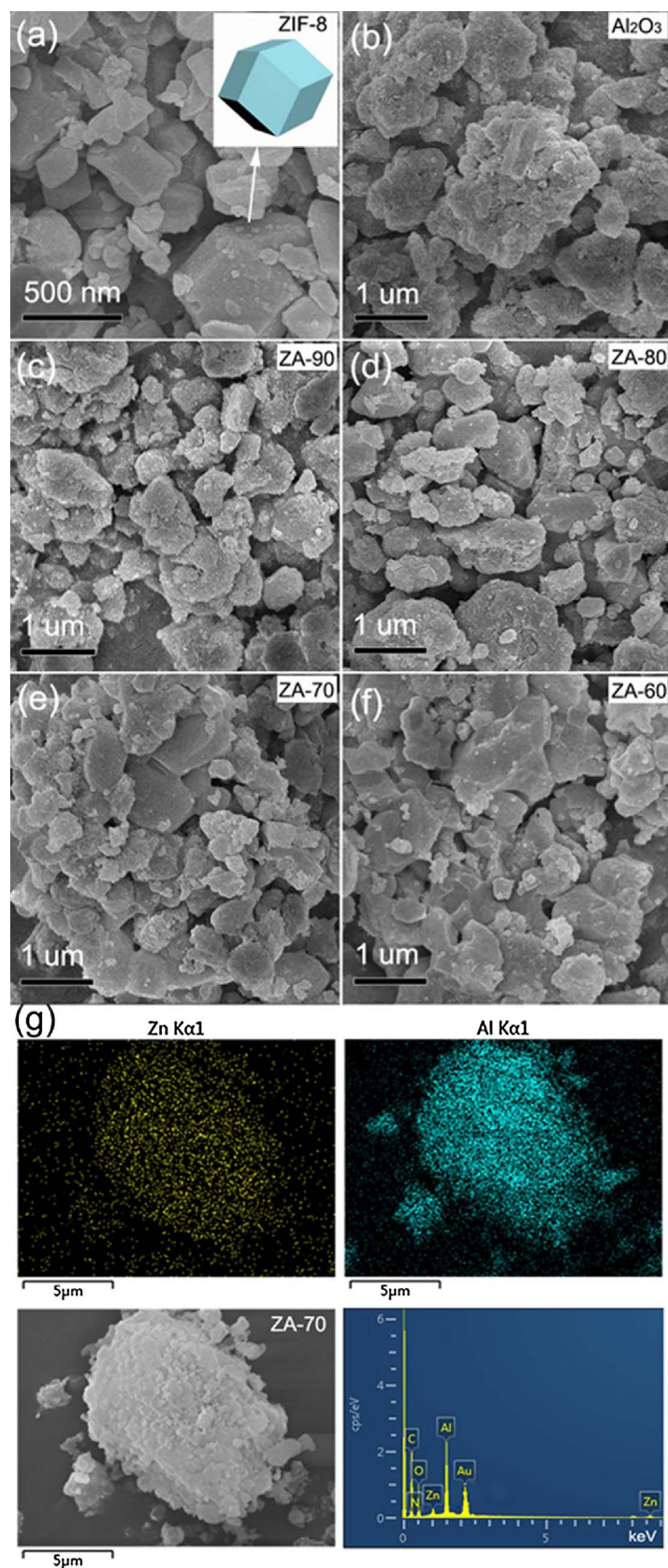


Fig. 4. SEM images of (a) ZIF-8, (b)  $\text{Al}_2\text{O}_3$ , (c) ZA-90, (d) ZA-80, (e) ZA-70, (f) ZA-60 and (g) elemental mapping images of ZA-70.

The elemental mapping results of ZA-70 (Fig. 4g) show that a homogeneous distribution of Al and Zn in ZIF-8@ $\text{Al}_2\text{O}_3$  composite, which also indicates that the  $\text{Al}_2\text{O}_3$  support makes the ZIF-8 well dispersed.

As shown in Fig. 5, the morphologies of  $\text{Al}_2\text{O}_3$ , ZIF-8, and ZA-n were

also characterized using TEM. ZIF-8 (Fig. 5b) have polygonal crystalline shapes which are consistent with the previous reports [37]. And the pristine  $\text{Al}_2\text{O}_3$  (Fig. 5a) nanorods were accumulated which lead to a mesoporous structure of particles. The composite components (ZIF-8

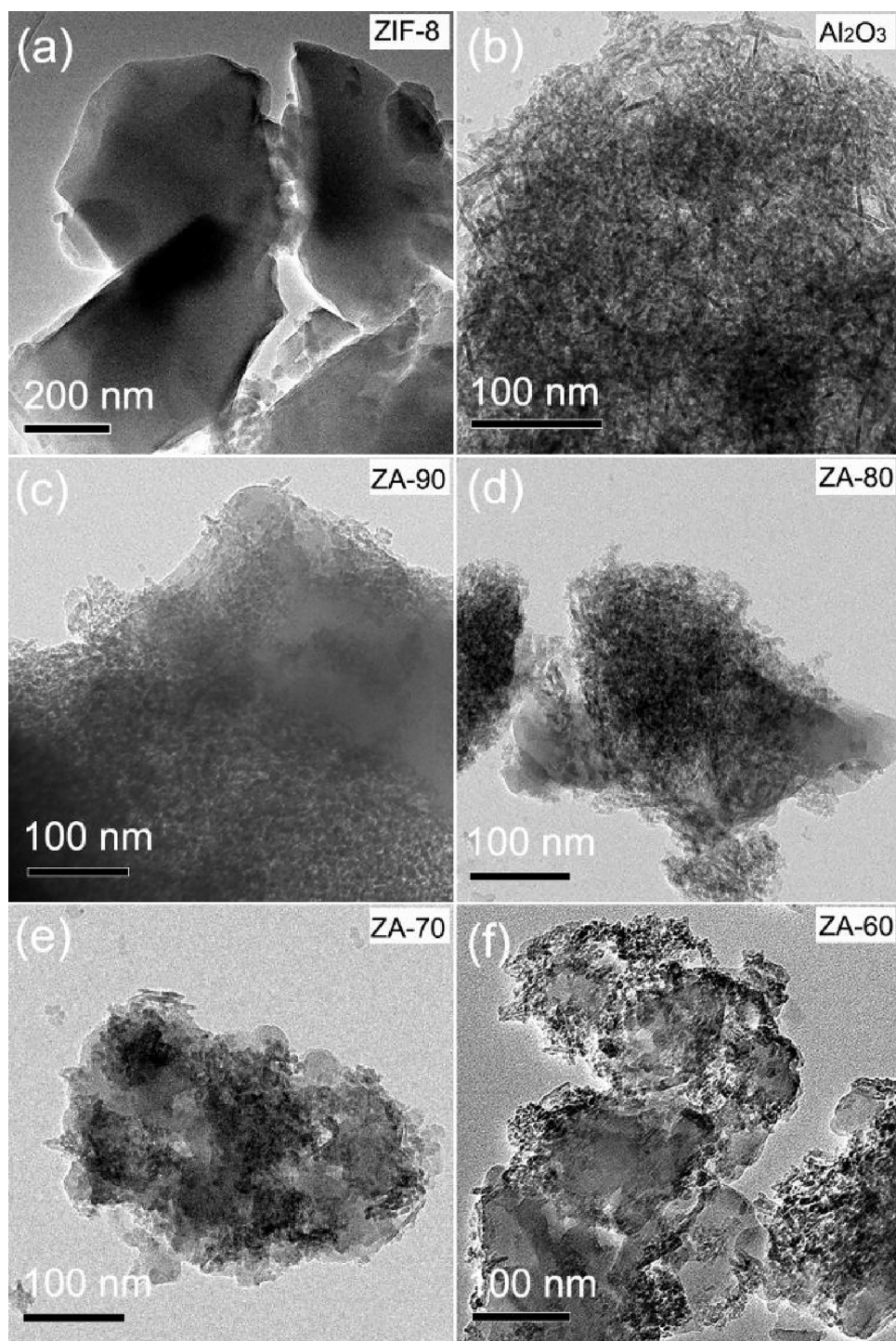


Fig. 5. TEM images of (a) ZIF-8, (b)  $\text{Al}_2\text{O}_3$ , (c) ZA-90, (d) ZA-80, (e) ZA-70 and (f) ZA-60.

with  $\text{Al}_2\text{O}_3$ ) are clearly visible in ZA-n (Fig. 5c–f), and almost no large particles of ZIF-8 was observed. Notably, ZIF-8 crystal tended to an inter-connected relationship with  $\text{Al}_2\text{O}_3$  matrix. That is likely ascribed to the ZIF-8 growth attached to  $\text{Al}_2\text{O}_3$  support by the abundant hydroxyl groups. As we know, ZnO nanoparticles can be well dispersed on  $\text{Al}_2\text{O}_3$  support through wetness impregnation method [1,39]. Interestingly, it seems that ZIF-8 linked and connected with each other, which is entirely different from the distribution of ZnO. That is because there was a destruction of ZnO reactant during solid reaction process, as illustrated in Fig. 2.

To analyze porous structure, nitrogen adsorption-desorption of ZA-n composites,  $\text{Al}_2\text{O}_3$ , and pure ZIF-8 were conducted (Fig. 6a). And the

porous structure parameters of samples are calculated from  $\text{N}_2$  sorption isotherms and listed in Table 1. Isotherm of  $\text{Al}_2\text{O}_3$  belongs to type IV classification, which shows a hysteresis loop typical of mesoporous materials due to the desorption process. ZIF-8 exhibits a type I isotherm characteristic of microporous. Interestingly, sorption isotherms of ZA-n composites can be viewed as a hybrid of type IV and type I, indicating the coexistence of micropore and mesopore. A sharp increase of nitrogen adsorption at low relative pressure ( $P/P_0$ ) region indicates the presence of micropore originating from ZIF-8. ZA-n samples show a full evolution of mesoporosity as the  $\text{Al}_2\text{O}_3$  ratio decreased from 90 to 60. Isotherms of ZA-n all exhibits a narrow hysteresis loop, and the loop becomes slightly narrower with the decrease of  $\text{Al}_2\text{O}_3$  ratio, which



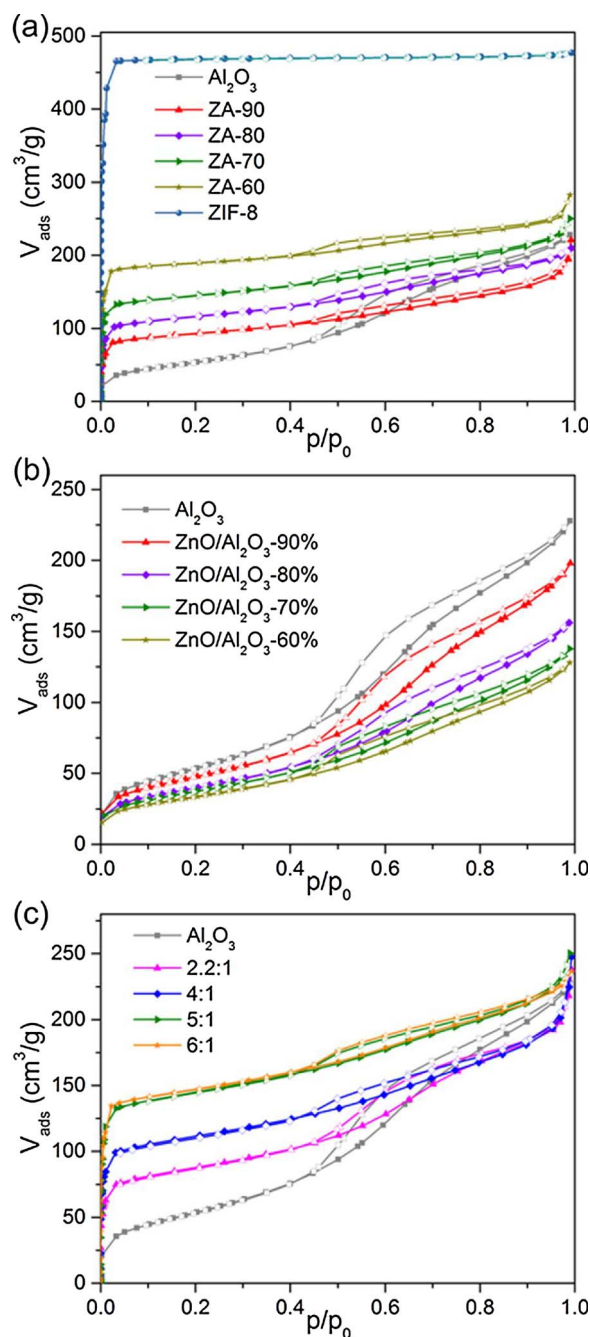


Fig. 6. Nitrogen adsorption-desorption isotherms of (a) ZIF-8@Al<sub>2</sub>O<sub>3</sub> with different proportion of Al<sub>2</sub>O<sub>3</sub>, (b) ZnO/Al<sub>2</sub>O<sub>3</sub> with different proportion of Al<sub>2</sub>O<sub>3</sub>, and (c) ZA-70 with different molar ratios (2-methylimidazole: ZnO).

indicates the decreasing of mesoporosity.

As listed in Table 1, textural parameters of pure ZIF-8 agree well with the published literature [36]. The BET surface area of ZA-n increased with the content of Al<sub>2</sub>O<sub>3</sub> decreasing, which was contributed by large microporosity of ZIF-8. For porous volume, microporous volume increased and mesoporous volume decreased with the content of Al<sub>2</sub>O<sub>3</sub> decreasing. These results indicate that a meso-microporous composite system has been achieved, which microporosity and/or content of ZIF-8 can easily be tuned by simply adjusting the loading of ZnO in precursor.

To analyze the relationship between ZnO and Al<sub>2</sub>O<sub>3</sub> support in ZnO/Al<sub>2</sub>O<sub>3</sub>-n, nitrogen adsorption-desorption of Al<sub>2</sub>O<sub>3</sub> and ZnO/Al<sub>2</sub>O<sub>3</sub>-n were also compared in Fig. 6b. All the samples show a hysteresis loop only characteristic of mesoporous. And the hysteresis loop becomes slightly narrower with the decrease of Al<sub>2</sub>O<sub>3</sub> ratio from 100% to 60%,

Table 1

Textural properties of Al<sub>2</sub>O<sub>3</sub>, ZA-n, ZnO/Al<sub>2</sub>O<sub>3</sub>-n, ZIF-8@Al<sub>2</sub>O<sub>3</sub>-70 prepared by 2-methylimidazole: ZnO feeding molar ratios 2.2: 1, 4: 1, 5: 1, and 6: 1, and ZIF-8.

Sample	S <sub>BET</sub> <sup>a</sup> (m <sup>2</sup> /g)	V <sub>total</sub> <sup>b</sup> (cm <sup>3</sup> /g)	V <sub>Micr</sub> <sup>c</sup> (cm <sup>3</sup> /g)	V <sub>Mes</sub> <sup>d</sup> (cm <sup>3</sup> /g)
Al <sub>2</sub> O <sub>3</sub>	199	0.35	–	0.35
ZA-90	448	0.35	0.18	0.17
ZA-80	455	0.38	0.21	0.17
ZA-70	588	0.39	0.23	0.14
ZA-60	631	0.36	0.24	0.11
ZnO/A-90 <sup>e</sup>	173	0.31	–	0.31
ZnO/A-80	145	0.24	–	0.24
ZnO/A-70	136	0.21	–	0.21
ZnO/A-60	123	0.20	–	0.20
ZA-2.2:1	327	0.35	0.20	0.15
ZA-4:1	436	0.36	0.19	0.15
ZA-6:1	588	0.37	0.24	0.13
ZA-8:1	601	0.36	0.24	0.12
ZIF-8	1660	0.59	0.58	0.01

S<sub>BET</sub><sup>a</sup> – BET surface area; V<sub>total</sub><sup>b</sup> – total pore volume obtained at P/P<sub>0</sub> = 0.99; V<sub>Micr</sub><sup>c</sup> – micropore volume calculated by t-plot analysis; V<sub>Mes</sub><sup>d</sup> – mesopore volume calculated by subtracting the V<sub>Micr</sub> from V<sub>total</sub>. ZnO/A-n<sup>e</sup> – ZnO/Al<sub>2</sub>O<sub>3</sub>-n, and n represent different mass percent of Al<sub>2</sub>O<sub>3</sub>.

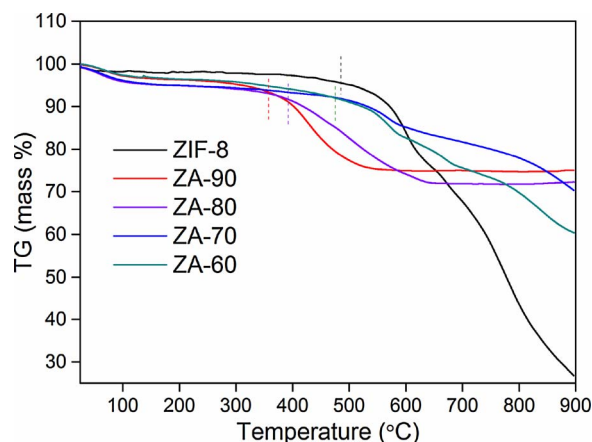


Fig. 7. TG analysis of ZIF-8, ZA-90, ZA-80, ZA-70 and ZA-60.

which indicates the decreasing of mesoporosity. Mesoporous volume of ZnO/Al<sub>2</sub>O<sub>3</sub>-90% to 60% listed in Table 1 also in agreement with this result. So we can conclude that ZnO was embedded in the pores of Al<sub>2</sub>O<sub>3</sub> support through the incipient wetness impregnation method.

The nitrogen sorption (Fig. 6c and Table 1) of products come from different molar ratio of starting materials (2-methylimidazole and ZnO) was also conducted besides XRD analysis (Fig. 3b). And the optimum 2-methylimidazole: ZnO feeding molar ratio is 5:1, which confirmed by a very similar N<sub>2</sub> sorption behaviors of the products prepared by molar ratios of 5: 1 and 6: 1. When the ratio is less than 5: 1, surface area and micropore volume were increased with the molar ratio increasing. That is because ZnO was in the pores of Al<sub>2</sub>O<sub>3</sub> support (analysed in Fig. 6b) and cannot fully contact and react with 2-methylimidazole when the molar ratio was less than 5: 1. This result further indicates the textural structure of ZIF-8@Al<sub>2</sub>O<sub>3</sub> composite, which is that ZIF-8 crystallites grow within Al<sub>2</sub>O<sub>3</sub> matrix.

TG analysis of ZIF-8 and ZA-n was conducted and the results were shown in Fig. 7. As can be seen, the weight loss curves trend of ZA-n with two stages are almost consistent with that of ZIF-8 during heating procedure. Firstly, a little weight loss of 3% for ZIF-8, and 4%–5% for ZA-n composites were detected at first stage, which is attributed to the removal of water and residual molecules (eg. 2-methylimidazole) in ZIF-8. Secondly, a sharp decrease in TG weight is caused by the decomposition of organic linker molecules and collapse of ZIF-8 structure. Compared with ZIF-8, the composite curve has a lower weight loss ratio at the content of thermal-stable Al<sub>2</sub>O<sub>3</sub>. Moreover, the decomposition

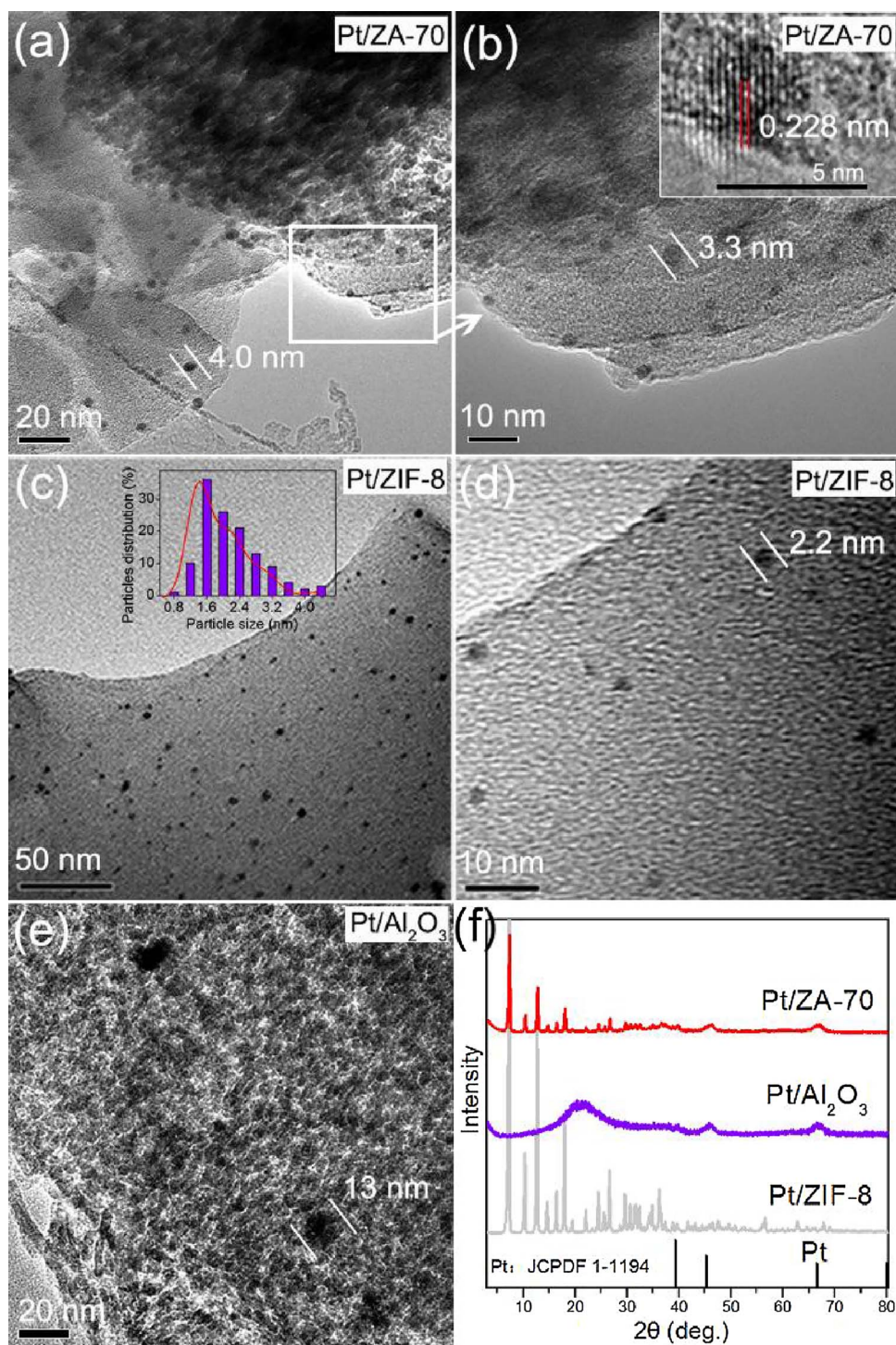


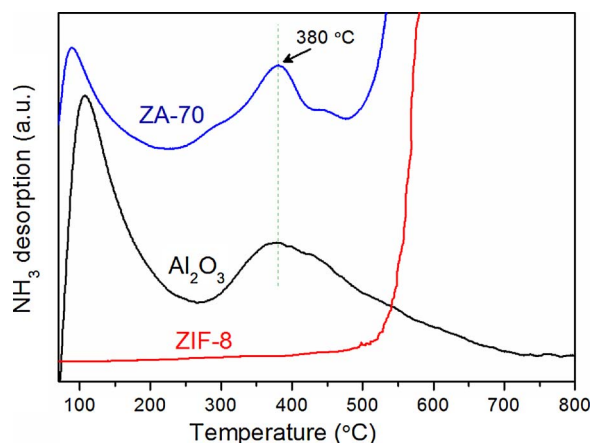
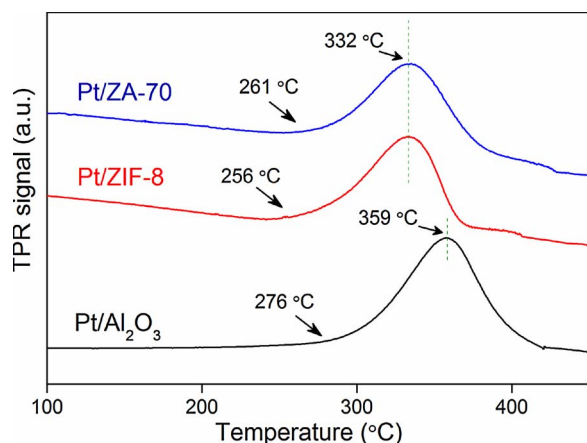
Fig. 8. TEM images of Pt/ZnO-Al<sub>2</sub>O<sub>3</sub> (a, b), Pt/ZIF-8 (c, d), Pt/Al<sub>2</sub>O<sub>3</sub> (e), and XRD (f).

temperature of ZA-70, and ZA-60 are started at around 475 °C which is close to the position of pristine ZIF-8 (485 °C). While composites ZA-90 and ZA-80 exhibit a lower decomposition temperature. That because an apparent activation energy for pure ZIF-8 is higher than that of composite materials, so a faster degradation reaction of ZIF-8 when forming a composite [40]. Especially, decomposition temperature of the composite is decreased with ZIF-8 content decreasing. The final residue, weighing 75.0%, 73.5%, 70.3%, 60.3% and 26.7% for ZA-90, ZA-80, ZA-70, ZA-60, and ZIF-8 respectively. The residue is ZnO, Al<sub>2</sub>O<sub>3</sub>, and also a little of carbon resulting from carbonization of organic ligand at second stage in the nitrogen atmosphere. It should be noted that because of an incomplete carbonization before 900 °C, so ZA-70, ZA-60,

and ZIF-8 have a relative higher weight loss than ZA-90, ZA-80. The above results show that ZIF-8 and ZA-70 can be used at lower than 475 °C, which will not damage its structure.

Porous γ-Al<sub>2</sub>O<sub>3</sub> is a frequently used support for hydrogenation catalysts. Hence, the ZA-n composite was used as support to prepare the catalyst and used for hydrogenation reaction. Pt/ZnO-Al<sub>2</sub>O<sub>3</sub> was synthesized using the incipient wetness impregnation method. Because ZIF-8 is a hydrophobic and not acid resistant MOFs, thus we chose an acetone solution of acetylacetonate platinum as an immersion solution, which as much as possible to make Pt enter into the pores of ZIF-8.

The particles size distribution of platinum nanoparticles of Pt/ZnO-Al<sub>2</sub>O<sub>3</sub> and Pt/ZIF-8 was examined by TEM (Fig. 8). Fig. 8a and b shows the

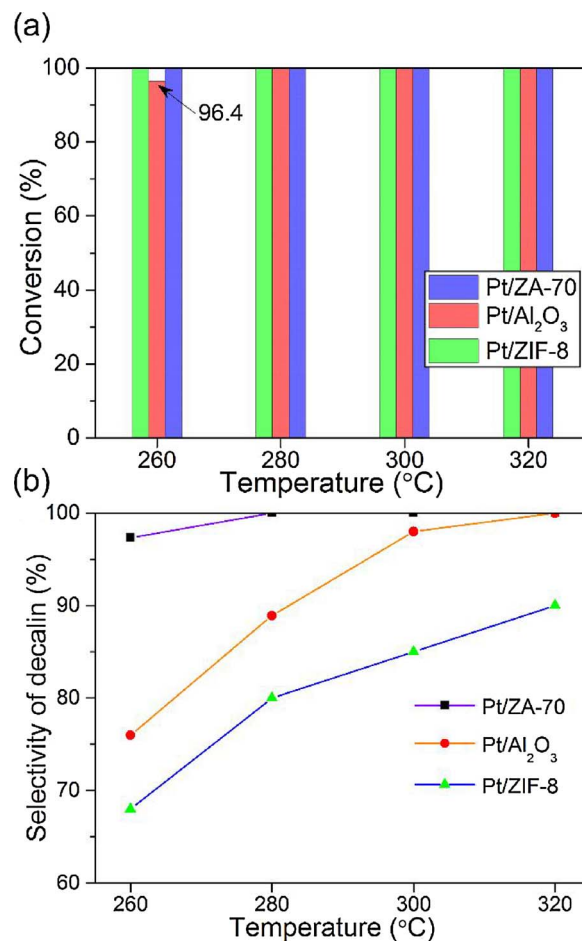
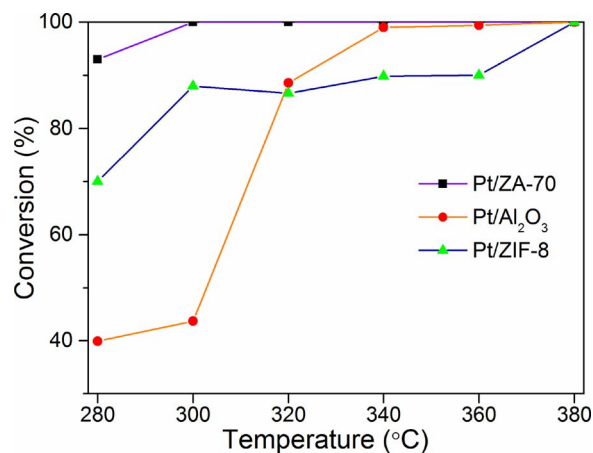
Fig. 9.  $\text{NH}_3$ -TPD of ZA-70, ZIF-8 and  $\text{Al}_2\text{O}_3$ .Fig. 10.  $\text{H}_2$ -TPR of Pt/ZA-70, Pt/ZIF-8 and Pt/ $\text{Al}_2\text{O}_3$ .

TEM images of Pt/ZA-70 catalysts, which indicates Pt nanoparticles in mean size of about 2–4 nm highly dispersed on ZIF-8 component of the composite. The HRTEM image inserted in Fig. 8b indicates that the interplanar spacing for the lattice fringe of is 0.227 nm, which correspond to the (111) lattice plane of the face-centered cubic (fcc) Pt structure [41]. The TEM images of Pt/ZIF-8 (Fig. 8c, d) indicate that highly dispersed Pt nanoparticles can be easily obtained by encapsulation of ZIF-8 using incipient wetness impregnation method. It also shows that acetone solution of acetylacetonate platinum is suitable for hydrophobic ZIF-8. However, a lot of large particles with size of about 13 nm appeared in Pt/ $\text{Al}_2\text{O}_3$  (Fig. 8e). Hence, ZIF-8 component can control a smaller size dispersed metal particles than  $\text{Al}_2\text{O}_3$ .

There is no observation of Pt characteristic peaks on Pt/ZA-70, Pt/ $\text{Al}_2\text{O}_3$ , and Pt/ZIF-8 (Fig. 8f), which should be due to a lower Pt loading (0.7 wt%) and small size of Pt nanoparticles. And the major peaks of Pt/ZA-70 and Pt/ZIF-8 are identical with that of ZIF-8, indicating that introduction of Pt can't destroy structure of ZIF-8.

In addition, X-ray photoelectron energy (XPS) was employed to investigate the catalytic architecture of Pt/ZA. The wide scanning spectra (Fig. S2a) indicates that Pt/ZA is mainly composed of C, N, O, Zn, Al and Pt elements. The spectrum of Pt 4f (Fig. S2b) displays one peak at 70.8 eV, corresponding to the  $\text{Pt}^0 4f_{5/2}$ , further demonstrating the successful encapsulation of Pt nanoparticles in support.

The acid properties of ZIF-8,  $\text{Al}_2\text{O}_3$  and ZA-70 were determined by  $\text{NH}_3$ -TPD (Fig. 9). Taking into account the thermal stability of ZIF-8, the temperature of  $\text{NH}_3$ -TPD for ZIF-8 and ZA-70 is up to 580 °C. Note that a sharp increase in  $\text{NH}_3$ -TPD curve start at about 480 °C due to collapse of ZIF-8 structure and decomposition of organic linker molecules, which is in accordance with the result of TG result (Fig. 7). The ZA-70

Fig. 11. Conversion and decalin selectivity over Pt/ZA-70, Pt/ $\text{Al}_2\text{O}_3$ , and Pt/ZIF-8 catalysts at different temperature.Fig. 12. Conversion of Pt/ZA-70, Pt/ $\text{Al}_2\text{O}_3$ , and Pt/ZIF-8 catalysts at different temperature in HDO reaction.

composite and ZIF-8 exhibited an identical desorption peak at 380 °C, which corresponding to intermediate acidic sites. While ZIF-8 sample has no peak in this range. Thus, there are a lot of medium acid sites on ZA-70, which comes from the contribution of  $\text{Al}_2\text{O}_3$  component and may play a critical role in activity and product selectivity during hydrogenation reaction [42].

The  $\text{H}_2$ -TPR results show that acetylacetonate platinum can be reduced by  $\text{H}_2$  at no more than 400 °C (Fig. 10). It is interesting to note that comparing with the Pt/ $\text{Al}_2\text{O}_3$ , the reduction peak of Pt/ZIF-8 and



**Table 2**Product selectivity of Pt/ZA-70, Pt/Al<sub>2</sub>O<sub>3</sub>, and Pt/ZIF-8 catalyst at different temperature in HDO reaction.

Products (%)	Catalysts	380 °C	360 °C	340 °C	320 °C	300 °C	280 °C
Hexadecane	Pt/ZA	91.2	15.5	8.4	1.4	0	0
	Pt/A	24.4	27.2	34.2	32.4	16.3	2.0
	Pt/Z	12.2	8.3	4.1	2.1	0	0
Hexadecene	Pt/ZA	0	70.1	65.2	60.3	42.0	36.4
	Pt/A	0	0	0	0	0	0
	Pt/Z	0	0	0	0	0	0
Pentadecane	Pt/ZA	3.6	2.5	1.8	1.3	0	0
	Pt/A	73.9	70.3	63.2	49.1	36.3	8.1
	Pt/Z	0	0	0	0	0	0
Hexadecanol	Pt/ZA	0	8.7	20.0	28.1	55.2	61.4
	Pt/A	0	0	0	0	0	0
	Pt/Z	73.2	70.3	67.1	66.3	30.9	15.2
Hexadecanal	Pt/ZA	0	0	0	0	0	0
	Pt/A	0	0	0	1.2	14.4	18.3
	Pt/Z	10.1	8.0	2.2	0	0	0
Palmitic acid	Pt/ZA	0	0	0	0	0	0
	Pt/A	0	0	0	11.4	29	66.3
	Pt/Z	0	7.8	19	18.6	57.8	65.1
Others	Pt/ZA	5.2	3.2	4.6	8.9	2.8	2.2
	Pt/A	1.7	2.5	2.6	5.9	4.0	5.3
	Pt/Z	4.5	5.6	7.6	13	11.3	19.7

Pt/ZA: Pt/ZA-70; Pt/A: Pt/Al<sub>2</sub>O<sub>3</sub>; Pt/Z: Pt/ZIF-8.

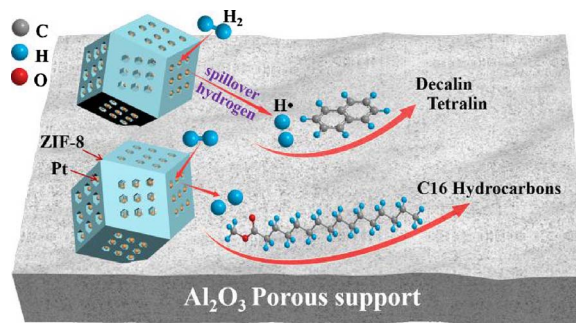
Pt/ZA-70 samples is apparently shifted to a lower temperature. The reduction peaks of Pt/ZA-70 and Pt/ZIF-8 are the same and 27 °C lower than that of Pt/Al<sub>2</sub>O<sub>3</sub>, implying that a smaller size and higher dispersion of Pt particles on ZIF-8 support. This result agrees well with TEM images in Fig. 8. In this paper, all the catalysts were reduced by H<sub>2</sub> at 350 °C for 1 h to insure the reduction of Pt<sup>2+</sup> before hydrogenation reaction.

Finally, it is noted that our approach doesn't involve any solvent and pretreatment. ZnO can achieve by wetness impregnation of zinc nitrate. So this approach may not be limited to ZIF-8@Al<sub>2</sub>O<sub>3</sub> materials, which holds promising to synthesize other porous support with ZIF-8.

### 3.2. Hydrogenation over Pt/ZA catalysts

Considering the properties of ZIF-8@Al<sub>2</sub>O<sub>3</sub>, 0.7 wt% Pt/ZA-70 catalyst was synthesized and its catalysis behavior was tested using hydrogenation reaction. In this work, Pt/ZA-70 was used as catalyst in HDA of naphthalene and HDO of methyl palmitate, respectively. For comparison, 0.7 wt% Pt/ZIF-8 and 0.7 wt% Pt/Al<sub>2</sub>O<sub>3</sub> were also tested under identical conditions.

Catalytic activity of Pt/ZA-n was firstly studied in HDA of naphthalene, in which decalin (including *trans*-decalin and *cis*-decalin) was the main product and tetralin was the by-product. As illustrated in Fig. 11a, all catalysts exhibit high conversion of naphthalene from 260

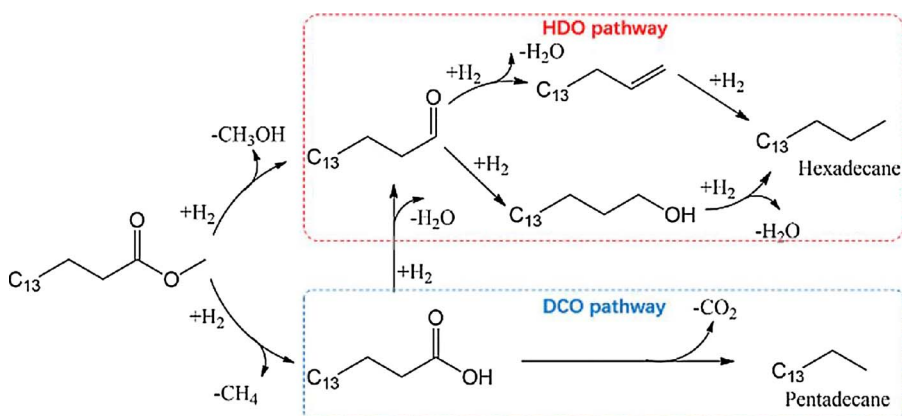
**Fig. 14.** Reaction pathway for the hydrodeoxygenation of Pt/ZA-70.

to 320 °C. Pt/Al<sub>2</sub>O<sub>3</sub> showed the lowest activity, which conversion of naphthalene was 96.4% at 260 °C. However, the conversion of naphthalene does not accurately reflect the activity of different catalysts. For HDA of naphthalene, the selectivity of decalin is most likely to accurately reflect the activity of the catalyst. Clearly, with ZIF-8 being incorporated in Al<sub>2</sub>O<sub>3</sub>, the decalin selectivity have been enhanced compared with its pristine components (ZIF-8 and Al<sub>2</sub>O<sub>3</sub>). Pt/ZA-70 showed the highest selectivity of decalin, which was 97.3% even at 260 °C. However, the selectivities of decalin for Pt/Al<sub>2</sub>O<sub>3</sub> and Pt/ZIF-8 at 260 °C were only 76% and 68%, respectively (Fig. 11b).

As previously published reports, the HDA of naphthalene relied on amounts of active sites as well as surface acid properties [43–45]. Tetralin is just an incompletely hydrogenated intermediate in HDA reaction. As analyzed by TEM (Fig. 8), Pt nanoparticles are dispersed uniformly in ZIF-8 component due to the “confinement effect” of MOFs, but its selectivity of decalin is the lowest. So the high decalin selectivity for Pt/ZA-70 catalyst generated not only from highly dispersed Pt nanoparticles, but also from acid property of Al<sub>2</sub>O<sub>3</sub> component.

For another type of hydrogenation reaction, HDO of methyl palmitate was also investigated for Pt/ZA-70, Pt/Al<sub>2</sub>O<sub>3</sub>, and Pt/ZIF-8, respectively. As shown in Fig. 12, Pt/ZA-70 shows the highest conversion of methyl palmitate, which conversion was 93% even at 280 °C. However, the conversion of methyl palmitate for Pt/Al<sub>2</sub>O<sub>3</sub> and Pt/ZIF-8 at 280 °C were only 39.9% and 70%, respectively. Product selectivities of various catalysts at different temperature in HDO reaction were listed in Table 2. Although Pt/ZIF-8 have a high conversion, but hydrocarbon selectivity was lower than 12.2% in measured temperature range. The major product of Pt/ZIF-8 is hexadecanol, which is an intermediate in HDO reaction.

According to our previous studies, there are two distinct pathways for deoxygenation for methyl palmitate. One is the cleavage of C–O bond and hydrogenation to produce hexadecane (HDO), and the other is the cleavage of C–C bond and decarbonylation to produce pentadecane (DCO) [46,47]. Acid properties of catalyst play a crucial role in hydrogenation activity and cracking activity for supported metal

**Fig. 13.** Reaction pathway for the HDO of methyl palmitate.

catalysts. High acidity of support can make the catalyst more selective towards C–C bond cleavage, and acidity also is important for the transformation of hexadecanol to hexadecane [48]. Detailed process of the two reaction pathways were concluded in Fig. 13.

It is noteworthy to mention that a significant change in product selectivity can be observed for Pt/ZA-70, Pt/Al<sub>2</sub>O<sub>3</sub>, and Pt/ZIF-8 (Table 2). As we can see, very few of hydrocarbon product was obtained on Pt/ZIF-8 in the whole temperature range. The dominate product is an intermediate hexadecanol, which might attribute to no acidity of ZIF-8. The catalyst cannot effectively adsorb the reactant molecules due to the absence of acidity, which will greatly reduce the catalytic activity. Conversely, major products obtained by catalyst Pt/ZA-70 above 320 °C were hexadecene and hexadecane. When reaction temperature reached 380 °C, hexadecane (up to 91.2%) becomes the main product, because higher temperature leads to the further hydrogenation of hexadecene to hexadecane. The excellent selectivity of hexadecane indicates that HDO almost was the only reaction pathway for Pt/ZA-70 catalyst. Whereas Pt/Al<sub>2</sub>O<sub>3</sub> yield dominant products were pentadecane and hexadecane, implying that DCO was the dominant reaction pathway, and also exists part of the HDO reaction pathway (Fig. 13).

The above results indicate that Pt/ZA-70 catalyst is more favorable for HDO pathway than DCO pathway, which might because most of the Pt nanoparticles were encapsulated in the pores of ZIF-8 and the acidic sites on the ZA-70 surface might play the role of adsorption reactant molecules. A possible bi-functional mechanism scheme was proposed to explain the effect of Pt/ZA-70 in hydrogenation, in which the ZA composite can simultaneously exert the advantages of ZIF-8 and Al<sub>2</sub>O<sub>3</sub> support. The micropores of ZIF-8 can increase the dispersion of Pt nanoparticles with smaller size. At the same time, the functional groups on the surface of Al<sub>2</sub>O<sub>3</sub> accompany with its mesoporous structure can improve the selectivity of the product. In this way, ZIF-8@Al<sub>2</sub>O<sub>3</sub> composite have significantly improved utilization efficiency of both ZIF-8 and Al<sub>2</sub>O<sub>3</sub> components. The detailed process of catalysis mechanism was illustrated in Fig. 14. H<sub>2</sub> molecules were adsorbed by ZIF-8 component with high surface area, and then were activated and dissociated to active H atoms by Pt nanoparticles. Subsequently, these activated H atoms were transferred to adjacent Al<sub>2</sub>O<sub>3</sub> component to react with reactant molecules due to the “spillover hydrogen effect” [49,50]. Simultaneously, reactant molecules can be adsorbed by acidic sites of Al<sub>2</sub>O<sub>3</sub>, which are free access to mesoporous of Al<sub>2</sub>O<sub>3</sub> and cannot enter into the pores of ZIF-8 limited by its microporous structure. Consequently, the composite system results in an increase of catalytic efficiency, and consequently improves catalytic activity and selectivity.

Notably, the confinement function of ZIF-8 component not only result in high dispersion of Pt nanoparticles, but also limit the growth of Pt nanoparticles under high reaction temperature. Additionally, the microporous structure and the non-acid sites of ZIF-8 can reduce the adsorption of reactant molecules in the pores, which might prevent the poisoning of Pt nanoparticles and coke formation. Hence, this kind of composite material might show completely different properties in various catalytic reaction.

#### 4. Conclusions

In conclusion, a new strategy was proposed to synthesize a novel composite of ZIF-8 embedded in porous Al<sub>2</sub>O<sub>3</sub> by a solvent-free solid-state method. In this method, supported zinc oxide was used as a precursor, and dimethyl imidazole was used as an organic ligand. Series of ZIF-8@Al<sub>2</sub>O<sub>3</sub> were synthesized arising from the crystal growth of ZIF-8 on Al<sub>2</sub>O<sub>3</sub> supports. BET results reveal that the distribution of micropores and mesopores can be controlled easily by simply adjusting the amount of zinc salt impregnated into porous Al<sub>2</sub>O<sub>3</sub> support. Additionally, the particle size of ZIF-8 crystal was significantly reduced due to its being embedded in porous support. It is also important to note that this approach doesn't involve any surfactants, solvent and pre-treatment process, which holds promising to synthesize other porous

support with ZIF-8. Pt/ZA-70 catalyst exhibits high activity and selectivity compared to Pt/Al<sub>2</sub>O<sub>3</sub> and Pt/ZIF-8 in HDA and HDO reactions. For HDA of naphthalene, the high decalin selectivity for Pt/ZA-70 composite generated not only from highly dispersed Pt nanoparticles, but also from acid property of Al<sub>2</sub>O<sub>3</sub> component. For HDO of methyl palmitate, the GC–MS results indicate that HDO almost the only reaction pathway for Pt/ZA-70 catalyst. Conversely, DCO was the dominant reaction pathway for Pt/Al<sub>2</sub>O<sub>3</sub> catalyst, which also exists part of the HDO reaction pathway. Finally, the experimental results show that there is a significant spillover hydrogen effect in the hydrogenation reaction for Pt/ZA-n catalyst. The special structure of the composite material might show completely different properties in various catalytic reaction.

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#### Appendix A. Supplementary data

Supplementary data associated with this article can be found, in the online version, at <https://doi.org/10.1016/j.apcatb.2018.01.022>.

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